

Alloyed monolayers of WX_2 of Al and Pt wires in Crystallographic phase: A DFT study

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ABSTRACT

We present density functional theory (DFT) based comprehensive study of two-dimensional (2D) alloyed monolayers of noble metals in hexagonal-phase within numerical atomic orbitals and plane wave basis sets methods. Positive phonon frequencies in the monolayers studied indicate that the metals are dynamically stable. The alloyed monolayers containing Pt show higher structural stability (binding energy and tensile strength) and exhibit metallic and ferromagnetic properties among the alloyed monolayers. Intriguingly, alloying monolayers with Al results in semiconducting behaviour, whereas alloyed WX_2 monolayers exhibit Dirac-cone-like properties at high symmetry points, in dielectric spectra for all of the structures investigated show anisotropic behaviour. For Pt-containing alloyed monolayers, tunnelling characteristics suggest an NDR zone. The considered alloyed monolayers may potentially be useful as building-blocks for the applications in nano and optoelectronics.

Keyword: Atomic wires, DFT, Alloyed Monolayers, Band structure, Mechanical properties, WX_2 .

1. Introduction:

Researchers have examined graphene, a two-dimensional (2D) honeycomb structure of carbon, in great detail [1–4], but it lacks a band gap. Many key applications in optics and transistor technology, however, need a band gap, prompting intense research efforts to functionalize it and manipulate the band gap. Layered transition metal dichalcogenides (LTMDs), such as WS_2 , have recently emerged as competitive materials for nanoelectronics due to unique features such as the presence of a band gap, temperature stability up to 1100 °C, and the lack of dangling bonds. The result was the successful development of a monolayer WS_2 -based field effect transistor recently. This has piqued people's curiosity in learning more about the unique features of different LTMD monolayers.

Although several features of bulk LTMDs have been examined before [6–12], it is only lately [13–14] that quantum confinement has been discovered to have a significant influence in the electrical and dielectric properties of multilayer TMDCs. A common example of a layered transition metal dichalcogenide family of materials is WX_2 ($X = S, Se, Te$). WX_2 bulk crystals have a hexagonal shape with space group $P6_3/mmc$, which corresponds to space group 194. They're built up with $X-M-X$ ($M=Mo, W$) units that are stacked van der Waals-bonded. A MX_2 monolayer is made up of two hexagonal planes of chalcogen (X) atoms and an intermediate hexagonal plane of transition metal (M) atoms that are coordinated by ionic-covalent interactions, with the X atoms arranged in a trigonal prismatic pattern as illustrated in Figure 1.

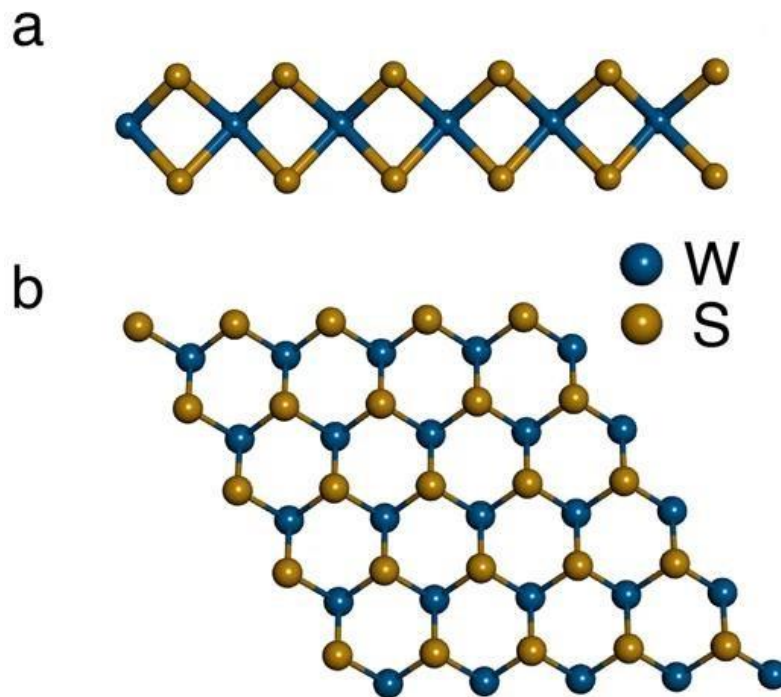


Figure 1: (a) Side view of WS₂ (b) Top View of WS₂

Novoselov et al. [15] developed a cleavage approach. They were able to obtain 6.5 Å thick honeycomb monolayers. Coleman et al. [16] also demonstrated that layered materials such as MoS₂, MoSe₂, MoTe₂, WS₂, TaSe₂, NbSe₂, NiTe₂, BN, and Te₂ may be exfoliated effectively into individual layers. Because of their ease of shearing, these materials are being used in lubricating applications [17]. Photocatalysis, optoelectronics, and photovoltaics are among the other uses of TMDCs that have been researched [18–22]. Optical spectroscopy has recently been used to investigate atomically thin MoS₂, revealing it to be a novel direct band gap semiconductor with a band gap of 0.7 eV to 2.0 eV [7–9]. MX₂ materials have a wide variety of band gaps, making them good candidates for device applications and maybe a viable alternative for Silicon (Si).

We set out to determine the electronic structure of sulphides, selenides, and tellurides of both Molybdenum (Mo) and Tungsten (W) after being inspired by recent investigations of the different features of monolayer WS₂ [23].

The lattice characteristics and electronic structure of monolayer MX₂ (M = Mo, W; S = S, Se, Te) described as 1H-MX₂ are reported in this study. Except for WTe₂, which has distorted octahedral coordination, all of the materials tested exhibit hexagonal symmetry (1H) with trigonal prismatic structure as shown in Figures 1 and 2, although we have only utilised hexagonal symmetry (1H) for direct comparison. By methodically lowering the thickness of materials from bulk to monolayer, we show how quantum confinement affects the electrical structure of distinct LTMDCs. The VASP technique was used to execute density functional theory (DFT) computations utilising ab-initio pseudopotential and numerical atomic orbitals (NAO's) basis sets inside the local density approximation (LDA). The prior electronic structure calculations only applied to bulk systems and the monolayer WX₂, however the results for the monolayers provided in this research are published.

Metal nanostructures' characteristics are known to be directly linked to their crystal structure. In bulk, noble metals have a variety of crystal forms; for example, Ag has two polymorphs, 9R-[24] and 4H-polytype [25]. Previous investigations [32–33] have shown the existence of hexagonal phase in one-dimensional (1D) hetero-nanostructures [26], thin films [26], nanorods [28–31], and nanowires of Silver. Marshall et al. [34] recently described a crystallisation route for the meta stable HCP phase in gold. Gold has also been synthesised in several nanostructures with HCP structure in the recent decade, including nanoparticles [35], nanowires [37], nanobelts or nanoribbons [38], nanoplates [6], nanosheets [39], and nanotubes [40]. The presence of gold nanoparticles in HCP frameworks makes understanding their characteristics intriguing from both a basic and technological standpoint.

Noble metal nanoplates and nanosheets have a wide range of uses, but their utilisation is limited due to limited reserves and expensive prices. Noble metals have been alloyed with earth plentiful metals to improve their qualities at a low cost [41]. The use of alloyed nanomaterials in which d-band vacancies of a Group 10 (Pt) metal are gradually filled by a Group 13 (Al) metal has piqued attention in the field of electronics [42–45]. It's worth noting that heterogeneous Al-Pt nanostructures with far superior characteristics than their pristine counterparts have been created for use in optics, electronics, protective coatings, bio-sensing, drug delivery, and catalysis [44–45]. Despite the fact that hexagonal structure is not an apparent choice for noble metal alloyed sheets, the above-mentioned recent findings have prompted it.

So far, quasi-1D batches of WX_2 -wires have been commonly generated in bulk inorganic synthesis by stabilising these atomic wires with different counter ions (such as alkali metals or iodine) [46–47] aligned and evenly spaced arrays of Pt and Al wires on Alloyed monomers, [49] free bundles of Pt and Al wires without counterions, Inside single-wall carbon nanotubes, WS_2 and WSe_2 wires were recently synthesised [50]. Individual MX_2 ($M = W$; $X = S, Se$), WSe_2 , and WS_2 alloy wires were also created by shaping their monolayer equivalents [48]. These atomic wires have self-adaptive connections to the monolayers and can withstand extreme mechanical deformation while remaining conductive. Individual wire twisting, axial kinking, and branching, as well as wellordered junction structures that join numerous atomic wires such as Pt and Al, have all been accomplished. Many types of subnanometer-width atomic wires are now possible because to developing manufacturing processes.

On the theoretical side, much work has gone into understanding the electrical, transport, and mechanical characteristics of metallic TMC wires, as well as investigating the implications of intercalation and heteroatom doping on their physical properties, particularly the likelihood of band gap formation opening [51]. Despite the aforementioned results, all prior experimental and theoretical investigations on metallic wires on molybdenum chalcogenides and tungsten chalcogenides as sandwich structures only concentrated on a few types of metallic wires. Because various 2D TMDs with a variety of structural phases and characteristics have been produced in the lab, [52] it's logical to wonder if additional transition metal elements besides Al and Pt may create stable 1D chalcogenide wires.

If that's the case, what would their geometrical structures be?

How do they compare to WX_2 -Al monolayers and WX_2 -Pt monolayers with atomic wires (Al and Pt), as well as their 2D TMD equivalents, in terms of electrical and mechanical properties?

Is it feasible to create 1D TMC wires with customised band topologies for a certain device?

To answer these exciting problems, we investigate atomic wires of transition metal chalcogenides in the stoichiometry of MX_2 ($X = S, Se, Te$; $M = 3d, 4d, \text{ and } 5d$ elements) for the first time. The electrical architectures of several types of wires are explored in order to screen 1D semiconductors and ferromagnets, respectively, for carrier mobility and Curie temperature. We go on to look into the mechanical characteristics of atomic wires, explicate the management of ferromagnetic exchange strength strain engineering, and offer a number of 1D flexible ferromagnetism with high Curie temperatures for spintronics.

2. Computational Methods

The Vienna Ab-initio Simulation Package (VASP) [53] was used to execute the computational computations for DFT-based first principle methods, employing the plane wave basis set with an energy cutoff of 500 eV and the projector augmented wave (PAW) [54] potentials to characterise the electron–ion interactions. For the exchange–correlation functional [56], Perdew, Burke, and Ernzerhof (PBE) parametrized the generalised gradient approximation (GGA) [55]. To minimise contact between the atomic wire and its duplicate, the unit cells of Al and Pt wires were periodic along the z axis and a vacuum space of 20 was added in the other two directions. A $1 \times 1 \times 8$, k-point grid based on the Monkhorst–Pack technique [57] was used to sample the Brillouin zone. With converging requirements of 10^{-6} eV for energy and 0.005 eV^{-1} for force, all 1D structures were thoroughly optimised for both ionic and cell degrees of freedom. The supercell approach was then used to calculate the phonon dispersions of all metal wires. We also estimated the electronic band structures of the semiconducting wires because the PBE functional is known to underestimate the band gap of a semiconductor.

3. Results and Discussion

3.1. Structure and Stability of 1D WX_2 -Atomic Wires

The atomic structures of WX_2 ($X = S, Se, Te$) wires, which were synthesised in the lab, are shown in Figures 1a and 1b. We created monolayers for the other 3d, 4d, and 5d transition metal chalcogenide wires based on existing metal wires. Because Al is a P-block element with a greater atomic size, traditional DFT approaches may be unable to accurately anticipate the geometrical and electrical architectures of Al atomic wires. Figures 1a and 1b show the optimal structures of all TMC atomic wires investigated. Metallic wires

made consisting of group III and VIII elements are generally strong conductors, however their 1D metallic wires have noticeable deformations and reduced symmetry. This is due to the group III and VIII transition metal atoms having fewer d electrons and a greater atomic radius. The M–M bonds are somewhat weak after contributing electrons to the S atoms; hence, structural deformation from the symmetric octahedral backbone can assist increase the M–S and M–M interactions.

Sulfide, selenide, and telluride monolayers with WX_2 monolayers have comparable structures for a given transition metal element M, with the M–X bond length rising as X moves from S to Se to Te. This paper presents thorough geometrical characteristics and bond overlap population study of several TMC atomic wires.

Table 1. Binding Energies (E_b), Formation Energies (E_{form}), and Optimized Lattice Constants (c) of alloyed-Atomic Wires

	E_b (eV)			E_{form} (eV)			c (Å)		
	S	Se	Te	S	Se	Te	S	Se	Te
Mo	5.92	5.57	5.25	0.52	0.43	0.22	4.36	4.45	4.59
W	6.11	5.77	5.44	0.33	0.25	0.03	4.37	4.46	4.57
Mn	4.09	3.71	3.34	-0.24	-0.36	-0.61	4.10	4.22	4.33
Fe	4.65	4.29		0.35	0.26		4.07	4.19	
Co	4.73	4.40	4.15	0.27	0.20	0.05	4.00	4.16	4.26
Ni	4.42	4.10	3.87	0.31	0.25	0.13	4.08	4.23	4.40
Pt	4.44	4.24	4.17	0.03	0.09	0.13	4.50	4.65	4.84

To characterise the dynamic stability of all the atomic wires, the phonon dispersions were estimated. In the whole phonon spectrum, all TMC wires are dynamically stable without imaginary frequency. We also confirmed that these metallic wires may be properly separated, meaning that they interact with one another via vdW interaction.

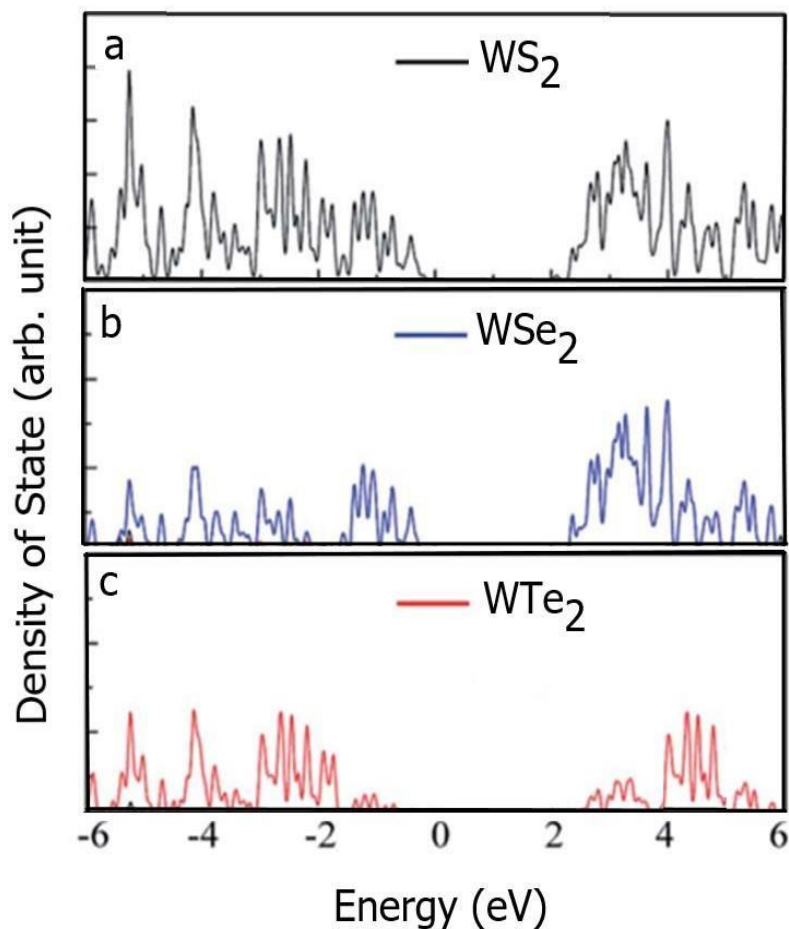


Figure 2: Density of state for (a) WS_2 (b) WSe_2 (c) WTe_2 respectively.

3.2. Structural properties

The computed values of the lattice constant, bond length, and binding energies of the examined systems are listed in Table 1. Our earlier research [58] described the structural characteristics of pure noble metal monolayers. The lattice constant values for the alloyed monolayers under investigation. The negative binding energy in each example (Table 1) indicates that all systems are energetically stable. The average binding energies of alloyed systems are higher than those of their pristine counterparts, indicating that alloyed monolayers are more stable than their pristine counterparts and that alloy production is encouraged VASP calculations show that the binding energies follow the $Al > Pt$ trend. It's worth noting that the values derived from two codes varied by around 6% (Table 1).

The Pt-containing alloyed monolayers are energetically more stable than the other alloyed monolayers investigated. All alloyed mono-layers tested were found to be dynamically stable.

3.3. Mechanical properties

When a crystal is squeezed or stretched from its equilibrium state, strain plays a vital role. It has the potential to alter device performance and can be used to improve device applications in NEMS and NOMS (nano electro-mechanical systems) (nano optomechanical systems). The value of ultimate tensile strain is equal to the greatest value of strain at which stress changes directly with strain, and the associated value of stress is equal to the system's ultimate tensile strength. The capacity of a system to bear load is referred to as its tensile strength.

Table 2 compares the values of ultimate tensile strain and strength derived from VASP for all investigated systems. It should be emphasised that the tensile strength of alloyed monolayers is comparable to that of their pure counterparts [58]. The tensile strength of Pt-containing systems is lower than pristine Pt monolayers but higher than other component atoms of alloyed monolayers. The maximum tensile strength among the examined systems is WS_2 -Pt, which varies as WS_2 -Pt $>$ WSe_2 Pt $>$ WS_2 -Al

>WTe₂-Pt >WSe₂-Al >WTe₂-Al. The greater bonding, as shown by the binding energies of the examined systems, is responsible for these developments.

3.4. Electronic and magnetic properties

The electronic band structure and related spin polarised total and partial density of states as a function of energy for WS₂-Pt, WS₂-Al, WSe₂-Pt, WSe₂-Al, WTe₂-Al, and WTe₂-Pt monolayers have been displayed in Fig. 2 to better understand the electronic features of alloyed monolayers. The band structures calculated using VASP are found to be comparable. The electronic band structures for several systems under investigation have been estimated along the Brillouinzone's Γ -M-K- Γ direction. All Pt containing alloyed monolayers (WS₂-Pt, WSe₂-Pt, and WTe₂-Pt) are metallic in nature, according to the electronic band structure, and their properties are fairly comparable to the electronic band structure of pure Pt monolayer described in earlier work [58]. Alloyed monolayers containing Al (WS₂-Al and WSe₂-Al) are semiconducting in nature. For the WS₂-Al monolayer, which sits between the M and K high symmetry points of Brillouinzone, an indirect band gap of 0.46 eV has been computed.

At the M high symmetry point, the WSe₂-Al monolayer has a direct bandgap of 0.88 eV. The band-gap values calculated by VASP differ by less than 2% [(Table 2) of Supplementary material]. In earlier research [59], alloyed nanowires of these noble metals exhibited a similar behaviour. Two conical intersections (Dirac-cone like behaviour) crossing the Fermi-level are seen in the WSe₂-Al monolayer, one from the conduction band at high symmetry point M and the other from the valence band at high symmetry point K. The behaviour of the WS₂-Al monolayer is comparable to that of virgin Pt and Al monolayers [58]. In case alloyed monolayers, the values of magnetic moment (b), tensile strength, and tensile strain are shown in Table 2.

For comparison	WS ₂ -Al	WSe ₂ -Al	WTe ₂ -Al	WS ₂ -Pt	WSe ₂ -Pt	WTe ₂ -Pt
Property	0.00	0.00	0.00	0.58	0.41	0.45
Magnetic Moment (μ_b)	(0.0)	(0.0)	(0.0)	(0.57)	(0.36)	(0.51)
Tensile strength (GPa)	1.77	1.97	1.49	2.49	1.86	2.64
Tensile strain (%)	(1.46)	(2.32)	(1.35)	(2.52)	(1.81)	(2.62)
(%)	16(14)	16(14)	16(14)	14(14)	16(14)	20(14)

We have also calculated the magnetic moment as given in Table 2 ($\mu_b = Q_{\uparrow} - Q_{\downarrow}$), where Q_{\uparrow} is the Mulliken charge population for the spin up and Q_{\downarrow} is the Mulliken charge population for the spin down states) Magnetic moments have also been estimated using the VASP algorithm, and the density of state graphs of the investigated systems are identical.

The density of states plots of all Pt containing alloyed monolayers (WS₂-Pt, WSe₂-Pt, and WTe₂-Pt) show that Pt, 5d orbitals around the Fermi-level contribute significantly to the density of states. Al, P orbitals contribute significantly to the density of states at the Fermi level in alloyed monolayers containing Al (WS₂-Al and WSe₂-Al) [Fig 2]. The density of states maps of the alloyed monolayers containing Pt (WS₂-Pt, WSe₂-Pt, and WTe₂-Pt) reveals that they are magnetic in nature. The magnitudes of the spin up and spin down states differ, with the spin up states contributing significantly, favouring ferromagnetism. The d-orbitals, notably the Pt 5d orbitals, provide a significant contribution to ferromagnetism. All of the other systems investigated are non-magnetic, with both spin up and spin down density of states equal in magnitude, resulting in no net magnetic moment.

Conclusion:

This research demonstrates that WS₂ nanotubes are both ultra-strong and elastic, setting them apart from other materials. A discrete number of atomistic flaws and the inherent strength of the W-S chemical bond determine the mechanical characteristics of WS₂ nanotubes. Individual nanotubes have therefore been proved to have perfect strength, demonstrating that defect-free structures are possible. Nonetheless, these high-strength, ductile materials are suitable for a wide range of applications, from nanocomposites to nanomachines.

In conclusion, we investigated the feasibility of 3d, 4d, and 5d transition metal and chalcogen (S, Se, and Te) elements forming 1D metal wires with well-defined geometrical shapes, inspired by synthetic atomic wires of Tungsten chalcogenides. We found atomic wires with adequate dynamic and thermal stabilities and various electronic band structures using high-throughput DFT simulations,

the majority of which could be manufactured from their bulk transition metal and chalcogen predecessors. Some of them may be air-stable. Our theoretical study suggests a family of 1D WS₂-wired materials with a diverse set of physical features.

They might open the door to discovering new states of matter through the quantum size effect and producing nanodevices with atomic accuracy.

In Summary, structural, electrical, magnetic, mechanical, dielectric, vibrational, and transport characteristics of ultra thin alloyed nanostructures were investigated using a comparative density functional theory (DFT)-based comprehensive analysis. Alloyed monolayers have similar lattice constant values to their pure counterparts. The average binding energy of these alloyed structures is higher than that of their pure counterparts, indicating that the alloyed structure is more energetically stable. The binding energy of Pt-containing alloyed monolayers is greater. All of the alloyed monolayers investigated had positive phonon frequencies for all modes at all high symmetry sites, indicating that the systems are dynamically stable. Metallic and ferromagnetic in nature, the Pt-containing alloyed monolayers have greater tensile strengths. The Al-containing alloyed monolayers have a band gap of 0.46 eV and 0.88 eV, respectively, and are semi-conducting. The infrared area contains the reflectance and absorption edges for alloyed monolayers. The alloyed monolayers under consideration might be beneficial in nanoelectronics, optoelectronics, magnetoelectronics, and spintronics.

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